



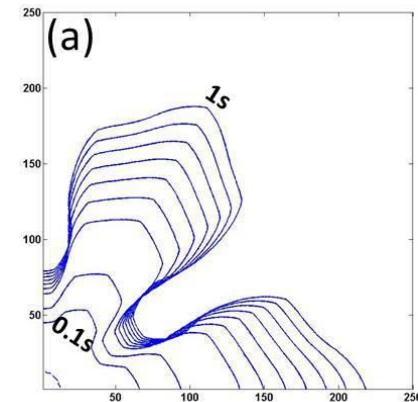
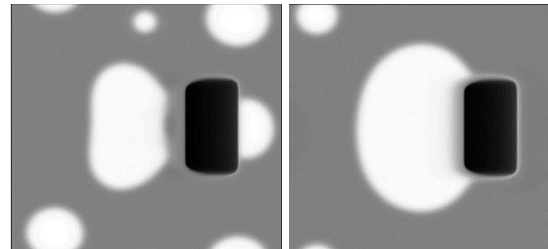
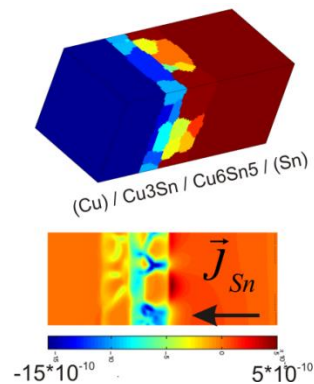
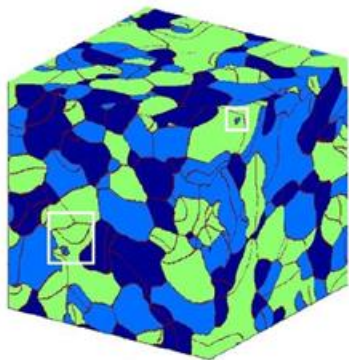
Modeling Factory Webinar

Phase Field modeling

Nele Moelans, KU Leuven

20 October 2016

- Dept. Materials Engineering, KU Leuven, Belgium
 - <http://www.mtm.kuleuven.be/English/>
- Solidification and Microstructure Simulation group
 - <https://www.mtm.kuleuven.be/Onderzoek/Semper/SolMicS>
- Phase-field method, multi-component alloys, microstructures



Outline

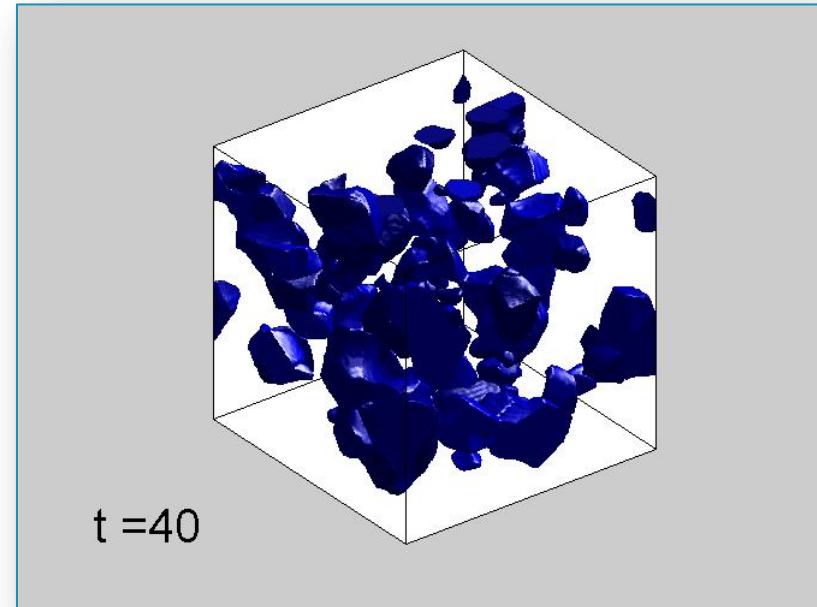
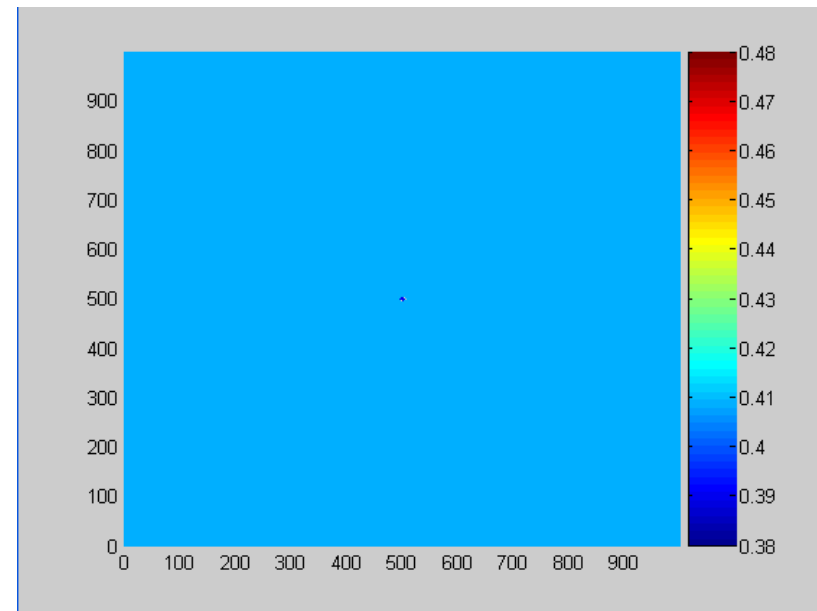
- Phase field method
- Examples
- Ideas on data management

Phase Field Method



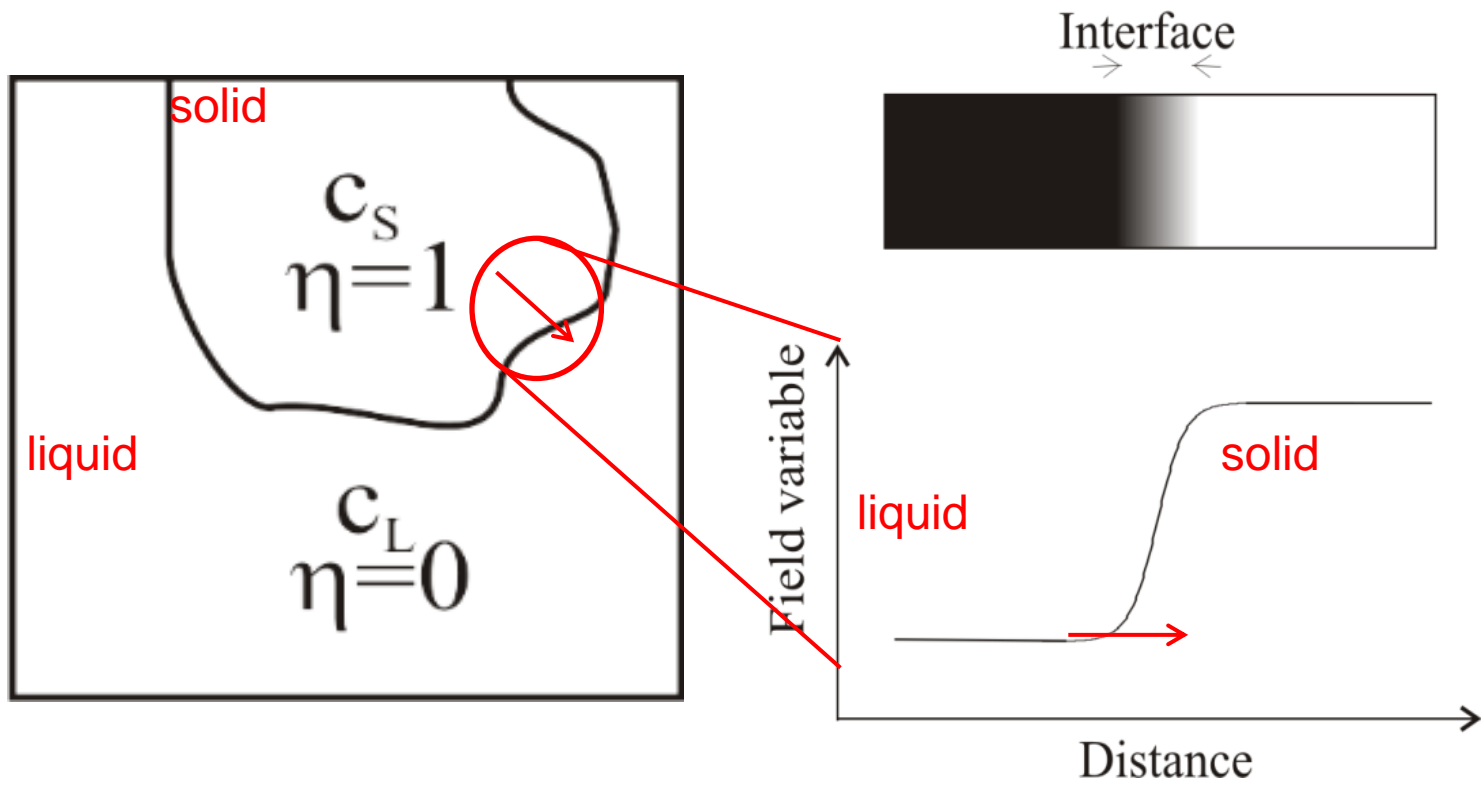
Phase-field method

- Microstructure evolution
 - Meso- & micro scale (5 nm - mm)
 - Thermodynamic principles
- Wide field of applications
 - Solidification, phase transitions, diffusion, grain growth, deformation, crack formation, ferro-electric/magnetic domains, nanoparticle, nanocrystals, thin films,
- Coupling with CALPHAD
→ multi-component, multi-phase



Microstructure representation

- Composition: $c_k(\vec{r}, t)$
• Structure, orientation: $\eta_i(\vec{r}, t)$ or $\phi_i(\vec{r}, t)$

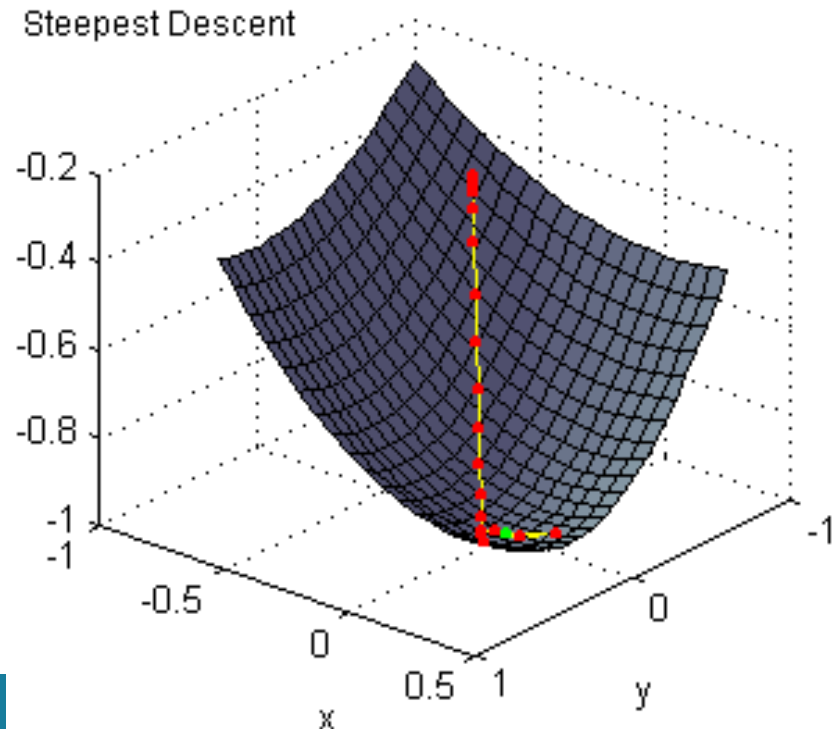


Evolution equations

- Non-conserved fields: $\eta_i(\vec{r}, t)$ or $\varphi_i(\vec{r}, t)$

$$\frac{\partial \varphi_i(\vec{r}, t)}{\partial t} = -L \frac{\partial F(\varphi_i, c_k)}{\partial \varphi_i(\vec{r}, t)} (+\vartheta_i(\vec{r}, t))$$

→ boundary motion



Evolution equations

- Concentration fields: $c_k(\vec{r}, t)$
 - Steepest descent + mass conservation

$$\frac{1}{V_m} \frac{\partial c_k(\vec{r}, t)}{\partial t} = \nabla M(\varphi_i, c_k) \nabla \tilde{\mu}_k \quad (+\theta_k(\vec{r}, t))$$

$$\tilde{\mu}_k = \frac{\partial F(\varphi_i, c_k)}{\partial c_k(\vec{r}, t)} = \mu_k - \mu_C$$

→ solute diffusion

Thermodynamic free energy

$$F = F_{bulk} + F_{interf}$$

$$= \int_V \left[f_0(c_k, \varphi_i) + \frac{\varepsilon_\varphi}{2} \sum_i (\nabla \varphi_i)^2 + \frac{\varepsilon_c}{2} \sum_k (\nabla c_k)^2 \right] dV$$

Homogeneous free energy

*chemical, elastic, ..., +
double/multi-well $g(\varphi_i)$*

Gradient free energy

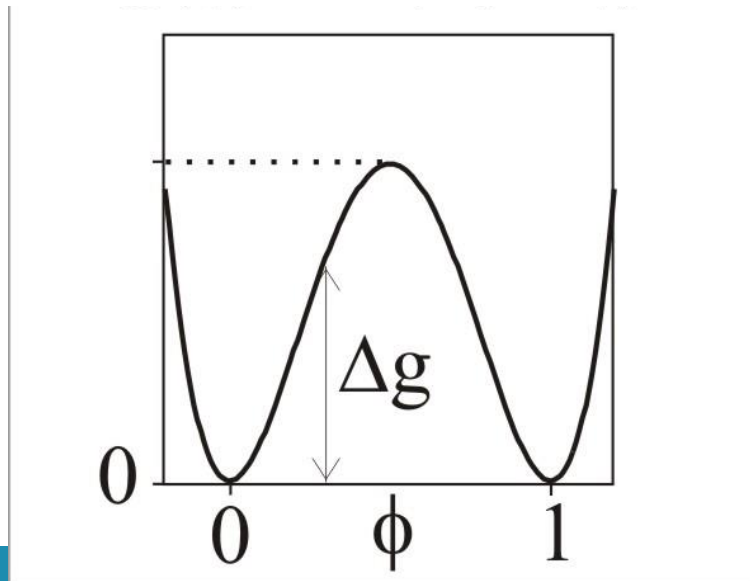
→ diffuse interfaces

Interface energy

$$F = F_{bulk} + F_{interf}$$
$$= \int_V \left[f_0(c_k, \varphi_i) + \frac{\varepsilon_\varphi}{2} \sum_i (\nabla \varphi_i)^2 + \frac{\varepsilon_c}{2} \sum_k (\nabla c_k)^2 \right] dV$$

Double well:

$$g(\varphi) = 16W\varphi^2(1 - \varphi)^2$$



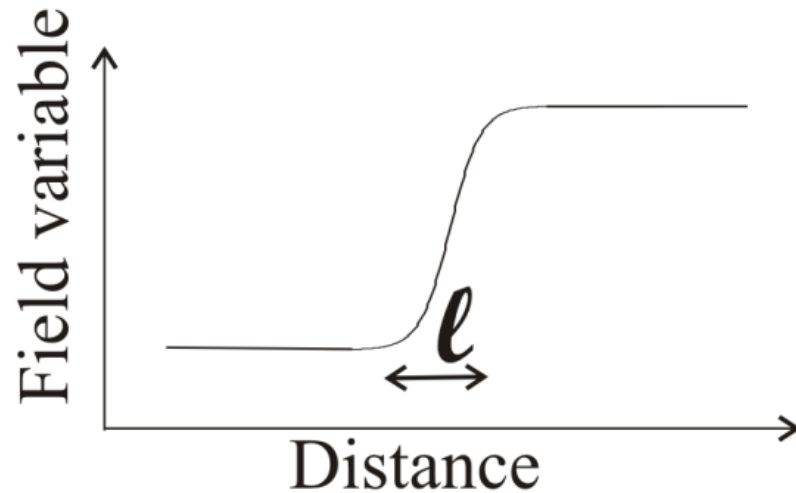
Interface energy:

$$\propto \sqrt{\varepsilon_\varphi}, \sqrt{\varepsilon_c}, \sqrt{(\Delta g)_{max}}$$

Interface width

$$F = F_{bulk} + F_{interf}$$

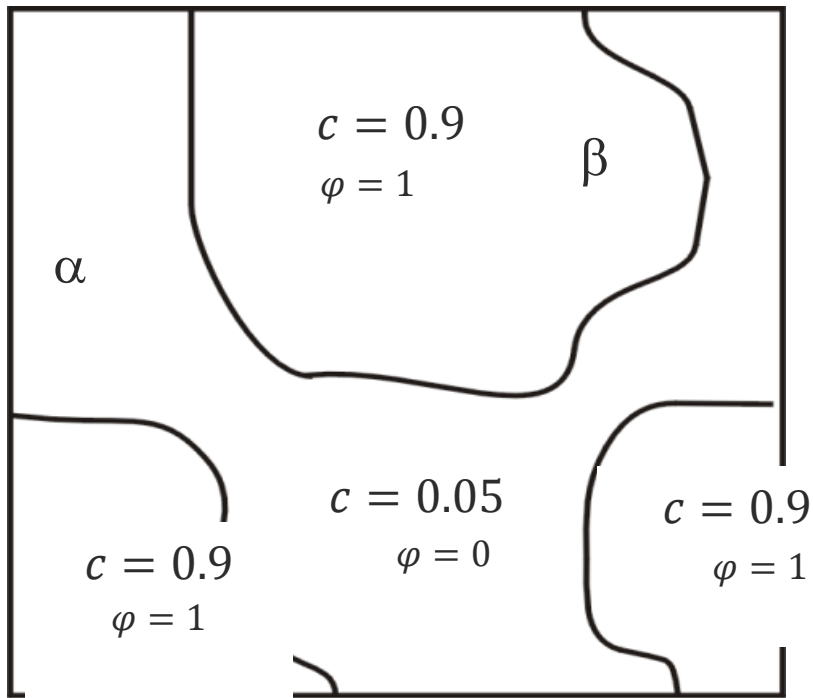
$$= \int_V \left[f_0(c_k, \varphi_i) + \frac{\varepsilon_\varphi}{2} \sum_i (\nabla \varphi_i)^2 + \frac{\varepsilon_c}{2} \sum_k (\nabla c_k)^2 \right] dV$$



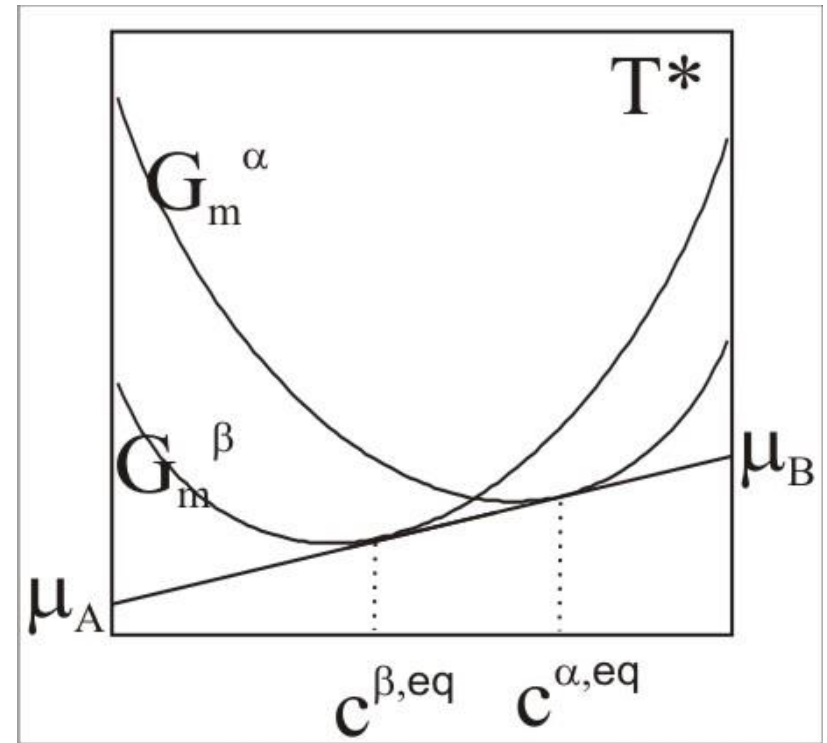
Interface width:

$$\propto \sqrt{\varepsilon_\varphi}, \sqrt{\varepsilon_c}, \sqrt{1/(\Delta g)_{max}}$$

Example: Binary 2-phase system



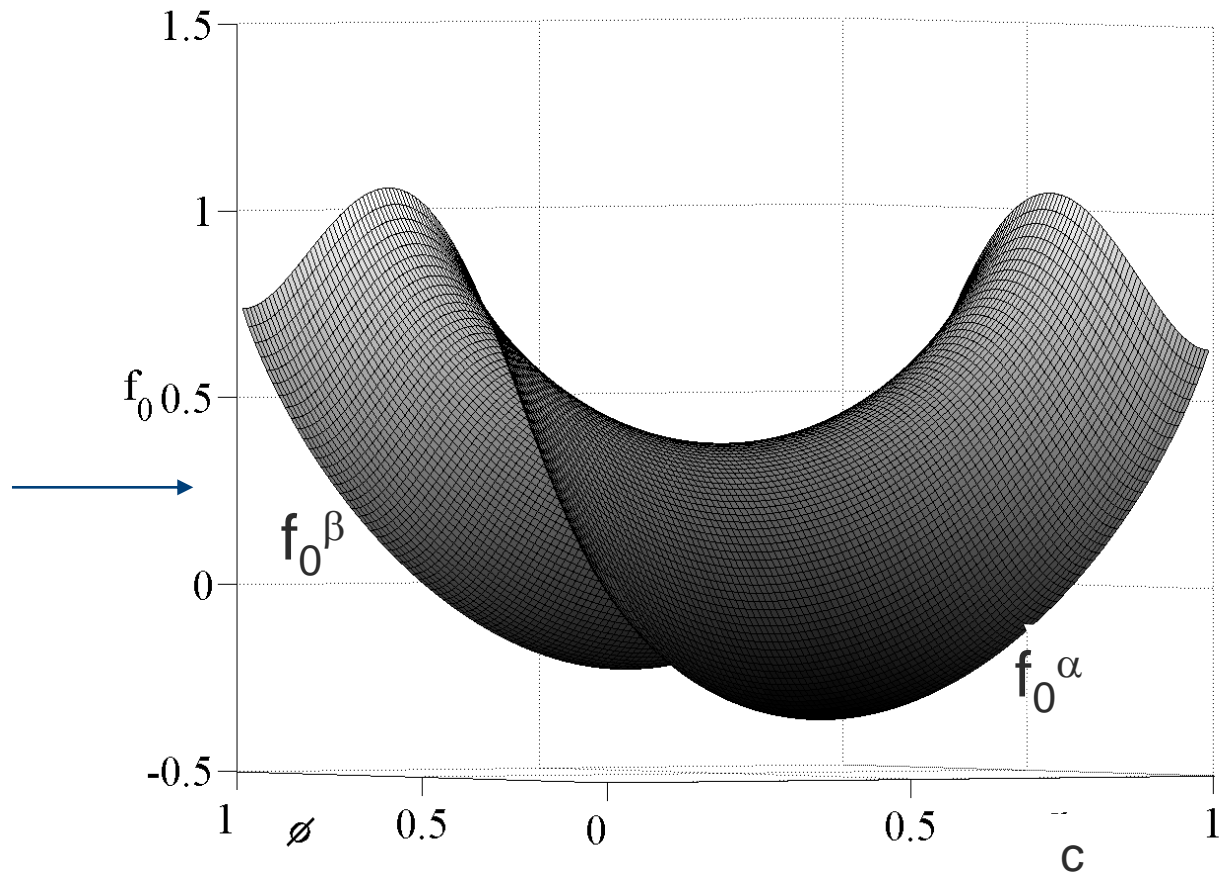
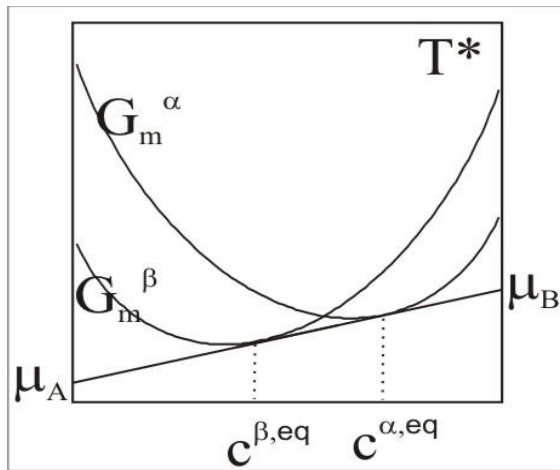
$\varphi(\vec{r}, t), c(\vec{r}, t)$



Gibbs energy α and β phase

$f_0(c, \varphi) ?$

Example: Binary 2-phase system



$$f^\alpha(c, T) = \frac{G_m^\alpha}{V_m},$$

$$f^\beta(c, T) = \frac{G_m^\beta}{V_m}$$

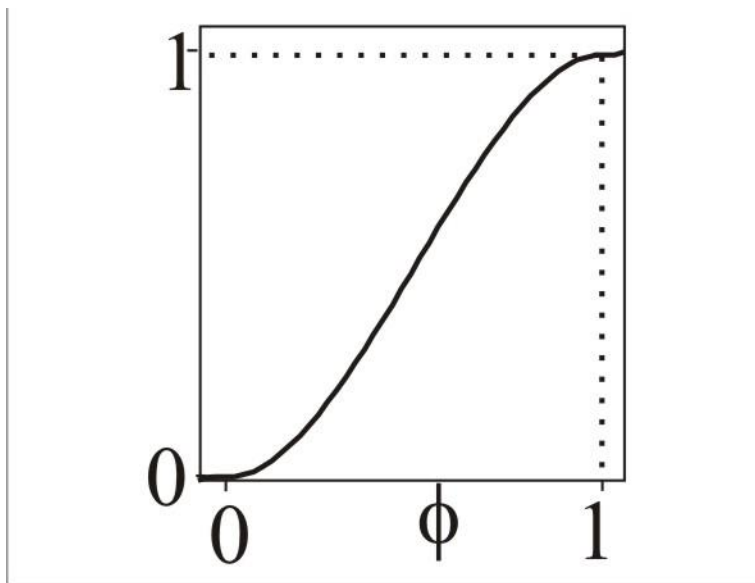
$$f_0 = h(\phi) f^\beta(c, T) + [1 - h(\phi)] f^\alpha(c, T) + \omega g(\phi)$$

Example: Binary 2-phase system

$$f_0 = h(\phi) f^\beta(c, T) + [1 - h(\phi)] f^\alpha(c, T) + \omega g(\phi)$$

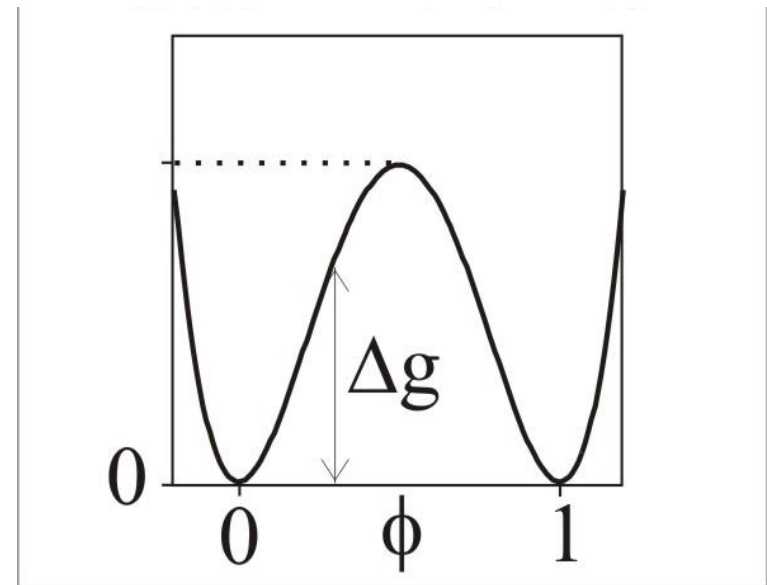
Interpolation:

$$h(\phi) = \phi^3(10 - 15\phi + 6\phi^2)$$



Double well:

$$g(\phi) = 16W\phi^2(1 - \phi)^2$$



Polycrystals/multi-phase

- Single phase: $\eta_1, \eta_2, \dots, \eta_i(\vec{r}, t), \dots, \eta_p$

within grain i

$$(\eta_1, \eta_2, \dots, \eta_i, \dots, \eta_p) = (0, 0, \dots, 1, \dots, 0)$$

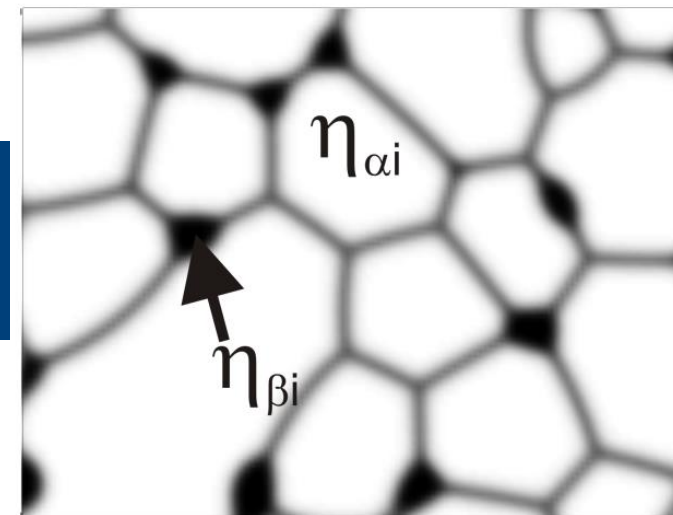
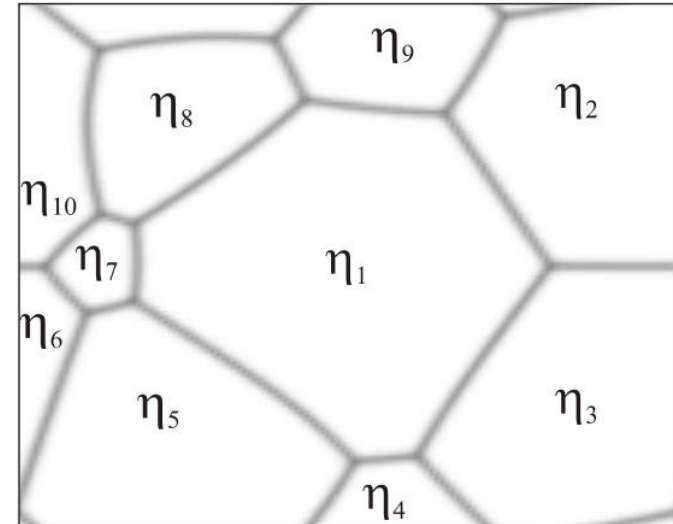
- Multi-phase

- Phases/grains

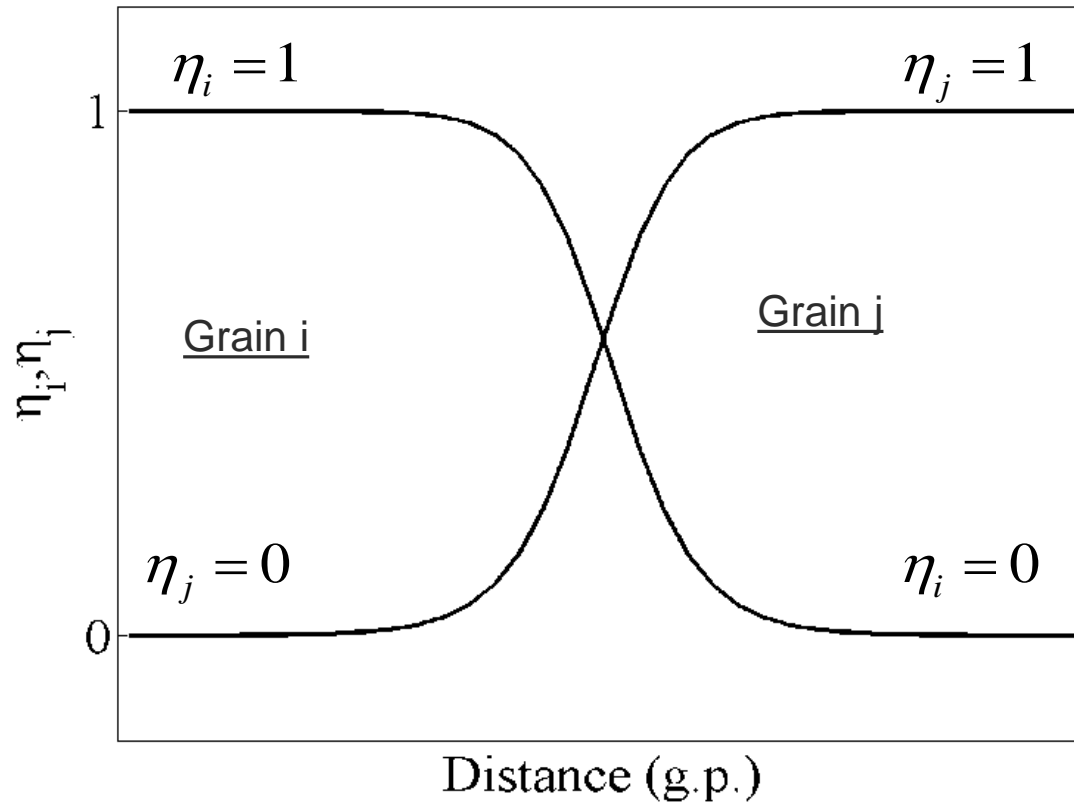
$$\eta_{\alpha 1}, \eta_{\alpha 2}, \dots, \eta_{\alpha i}(\vec{r}, t), \dots, \eta_{\beta 1}, \eta_{\beta 2}, \dots, \eta_p$$

- Composition

$$c_A, c_B(\vec{r}, t), \dots, c_{C-1}$$



Polycrystals/multi-phase



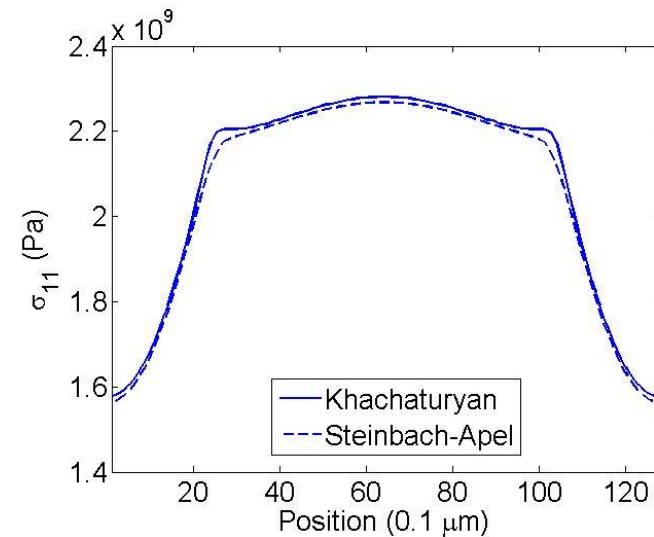
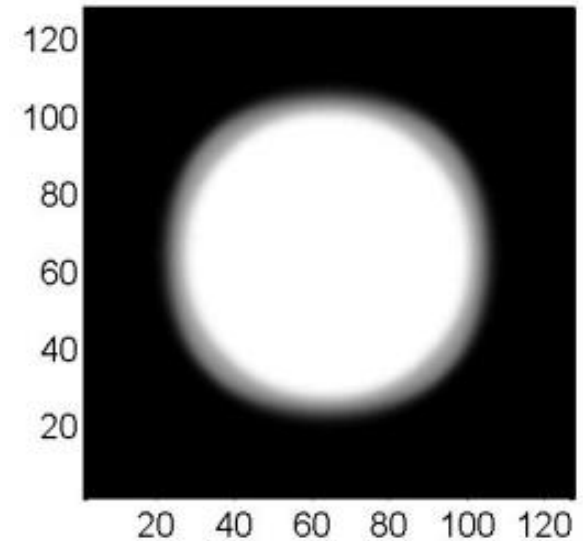
Other driving forces

$$F = F_{chem} + F_{elast} + F_{ferromag} + \dots$$

- Micro-elasticity theory

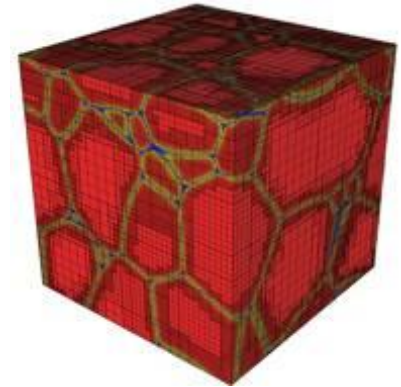
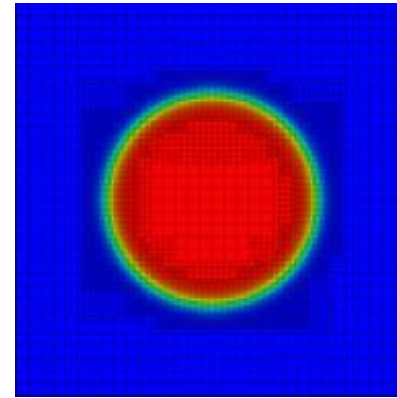
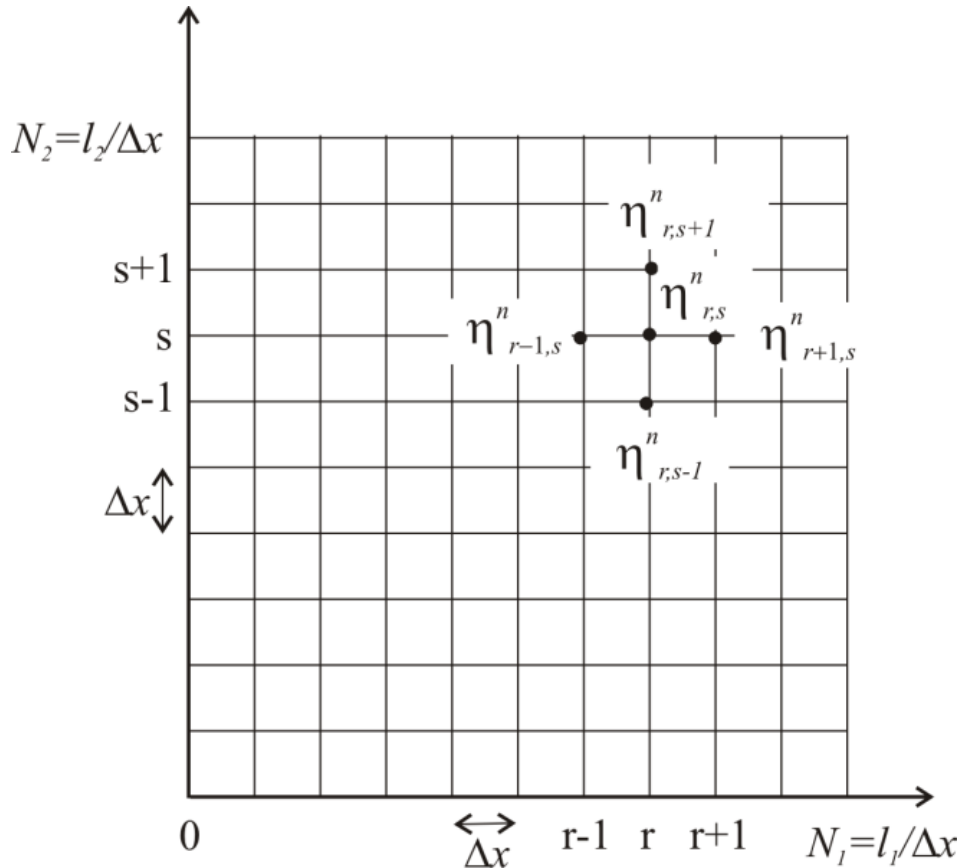
$$F_{elast} = \frac{1}{2} \int_V C_{ijkl}(\eta, x) \varepsilon_{ij}^{el}(\eta, x) \varepsilon_{kl}^{el}(\eta, x) dV$$

- Transformation, applied and thermal strains/stresses
- Solid-solid transformations

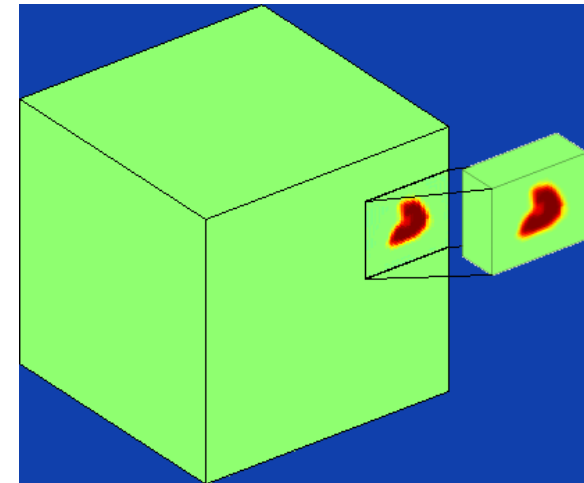


Numerical implementation

- Finite differences, finite elements, fourier-spectral



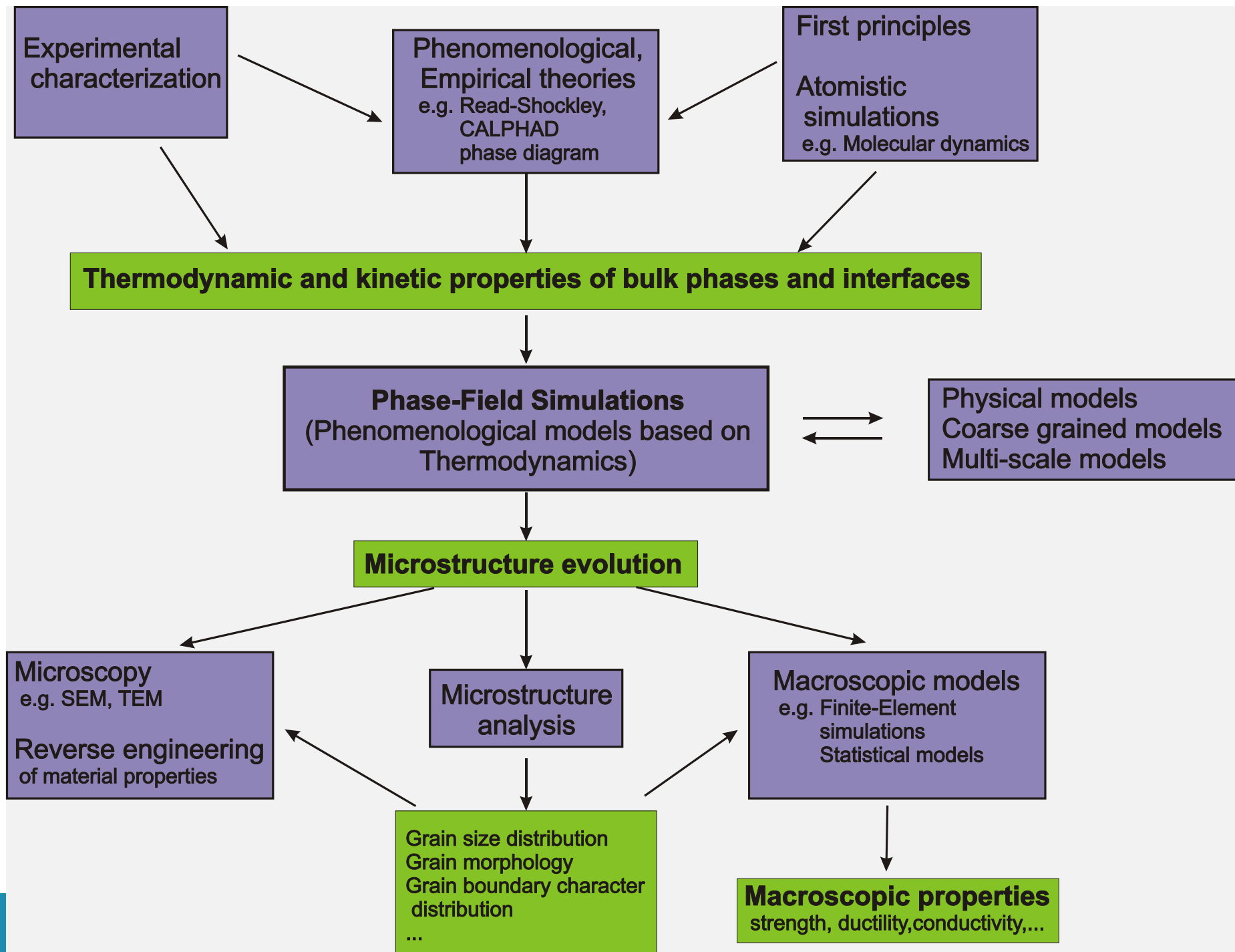
Adaptive meshing



Sparse data structures

Parameter assessment

- Different kinds of input data
 - Phase stabilities, phase diagram
 - CALPHAD ← experiments, DFT
 - Interfacial energy and mobility
 - MD, DFT, experimental
 - Elastic properties, crystal structure, lattice parameters
 - ab-initio, experimental
 - Diffusion mobilities
 - DICTRA ← experimental, MC ← DFT
- Orientation and composition dependence

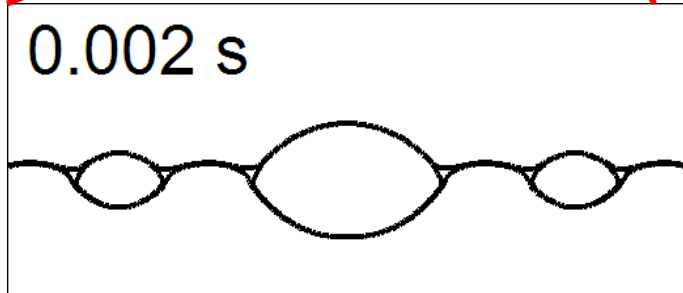
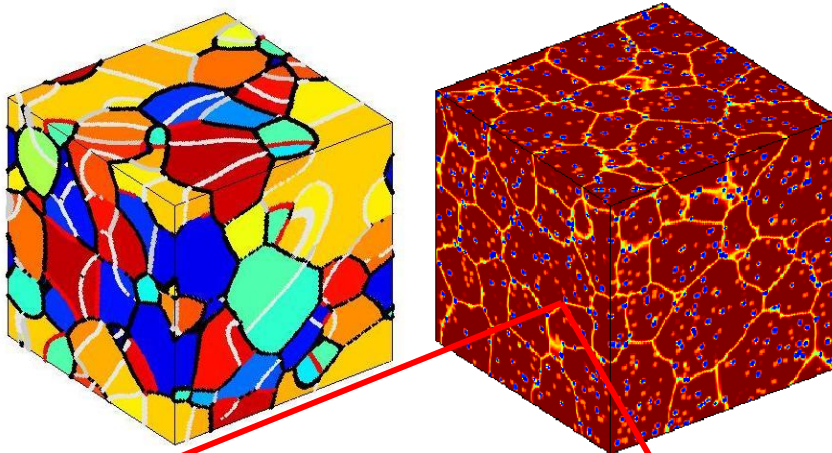


Examples

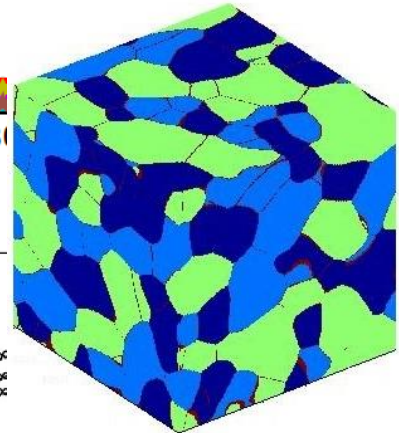
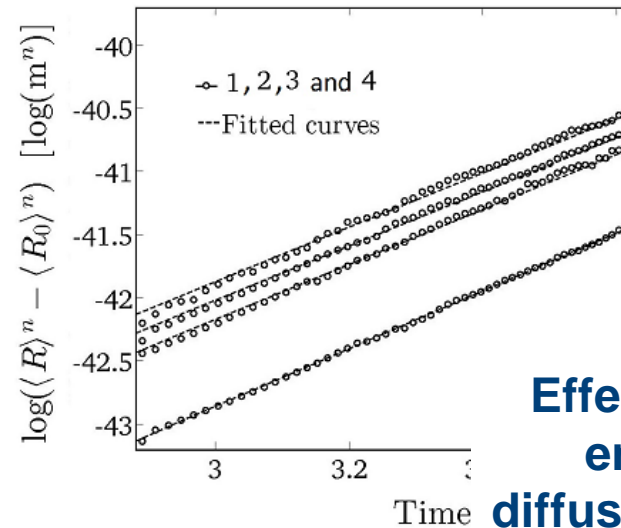
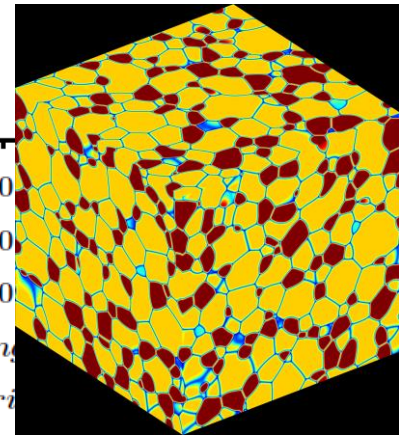
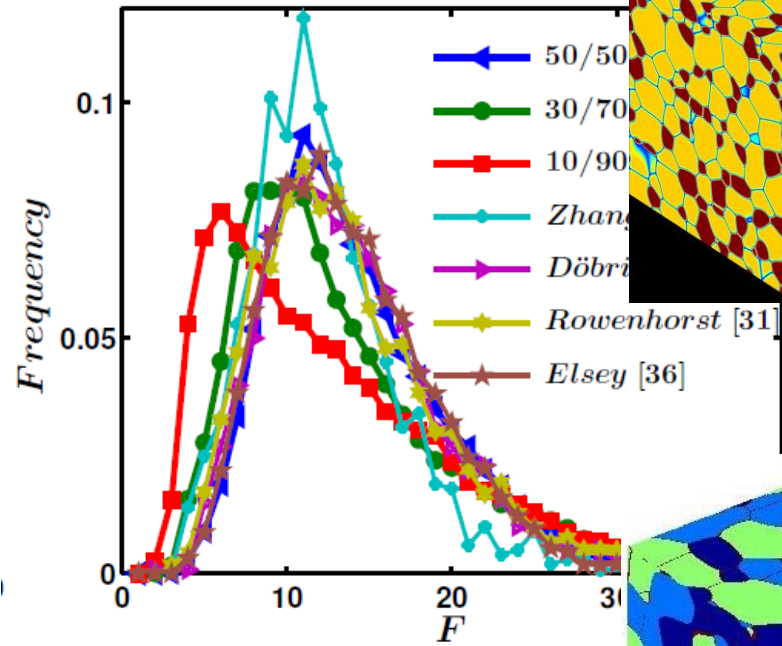
- Polycrystals
- IMC growth at material interfaces
- Crystallization in oxide systems
- Wetting in metal-oxide systems

Polycrystals

Effect of anisotropy and precipitates



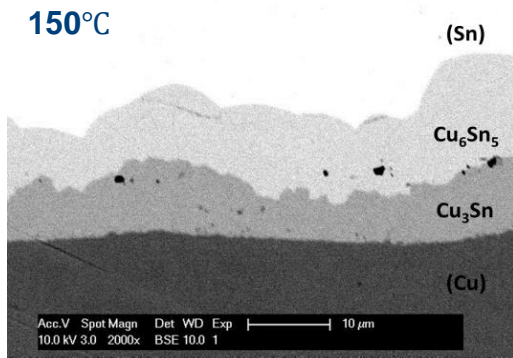
Multi-phase materials



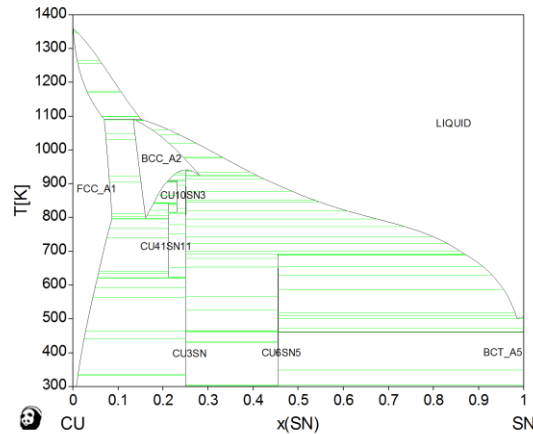
Effect of interface energies and diffusion coefficients

Modeling IMC growth

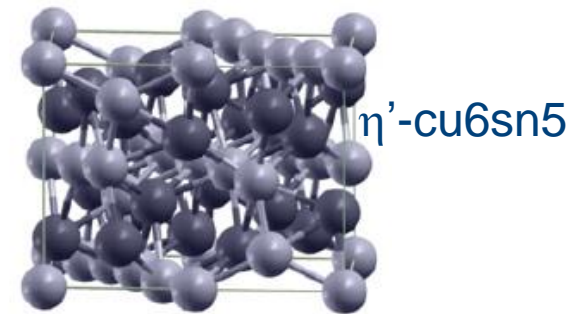
Experiments



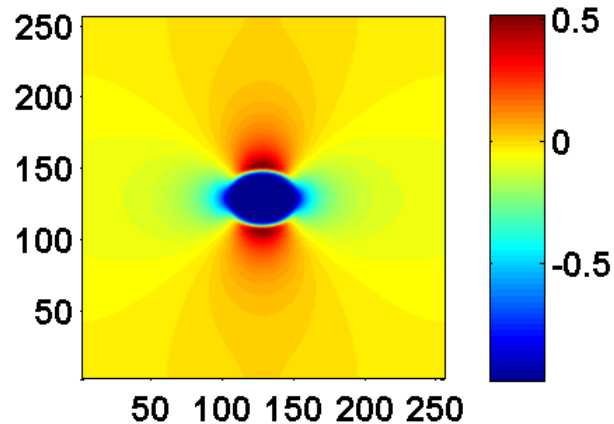
CALPHAD/DICTRA



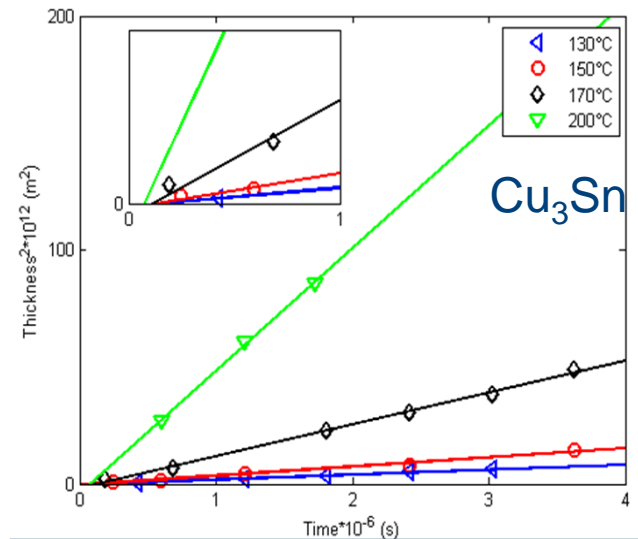
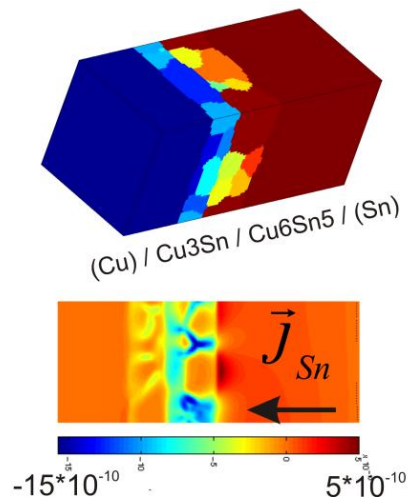
Ab initio calculations



Micro-elasticity and plasticity model



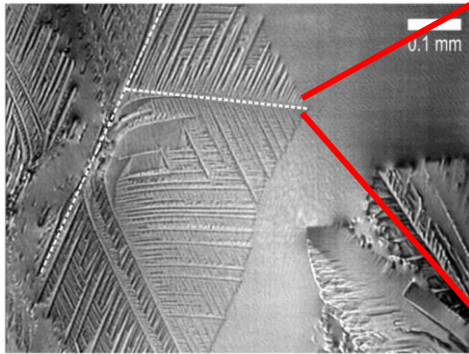
Phase field simulation



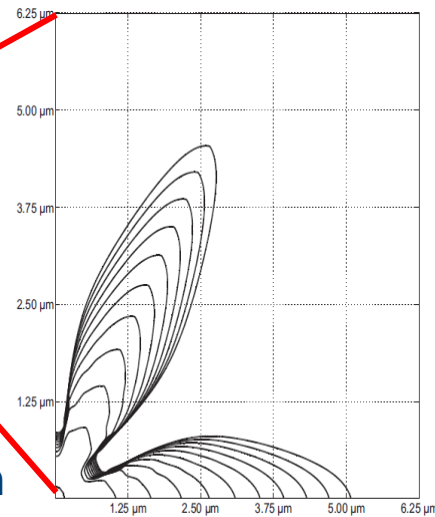
Analysis: growth rates, morphology, diffusion paths, solubilities, diffusion coefficients

Crystallization in oxide melts

Wollastonite growth from
 $\text{CaO-Al}_2\text{O}_3\text{-SiO}_2$ melts

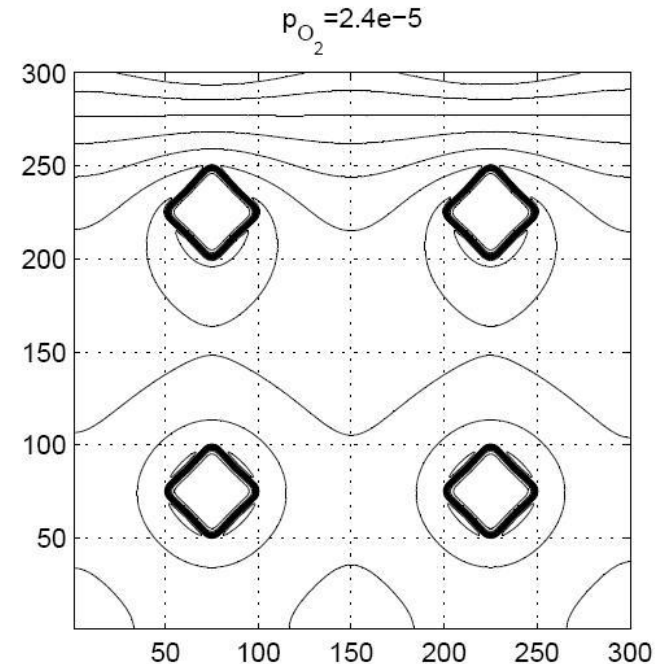


In-situ CSLM observation



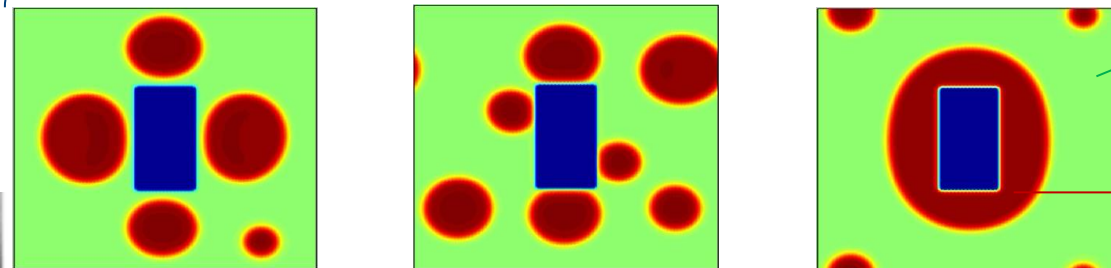
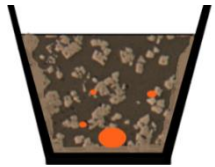
Simulation

Magnetite growth in oxidizing
 $\text{FeO-Fe}_2\text{O}_3\text{-SiO}_2$ system



Wetting in metal-oxide systems

Sticking metal droplets



Effect of interfacial energies

$$\begin{aligned} \gamma_{S,LO} &= 1.2641\text{N/m} \\ \gamma_{S,LM} &= 4.0093\text{N/m} \end{aligned}$$

$$\begin{aligned} \gamma_{S,LO} &= 1.2641\text{N/m} \\ \gamma_{S,LM} &= 2.0260\text{N/m} \end{aligned}$$

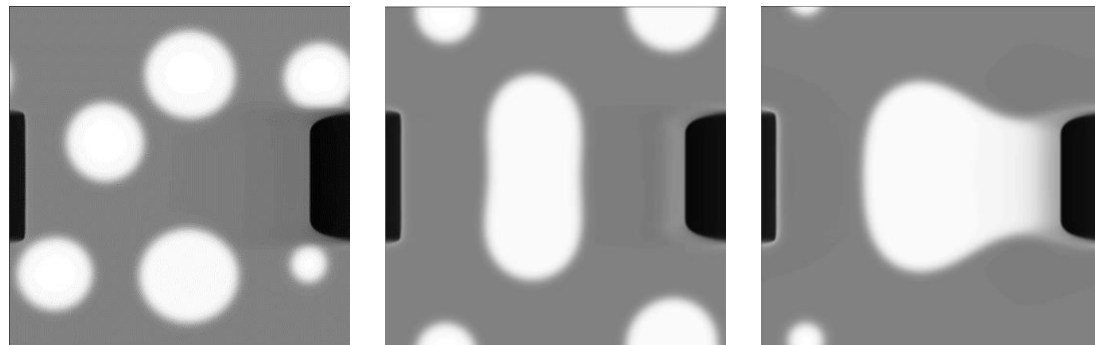
$$\begin{aligned} \gamma_{S,LO} &= 1.6318\text{N/m} \\ \gamma_{S,LM} &= 1.3325\text{N/m} \end{aligned}$$

Oxide slag

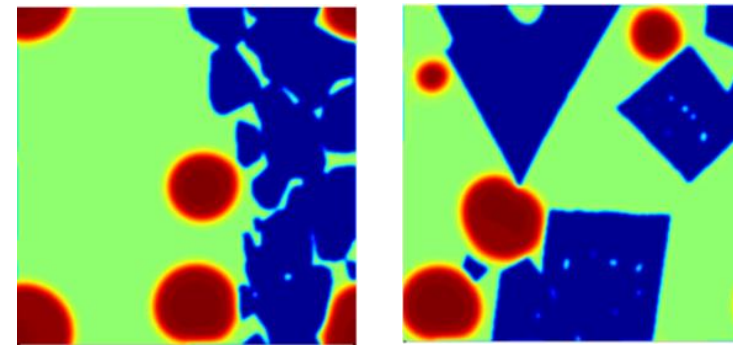
Metal droplets

Spinel particle

Effect rigid body motion



Realistic spinel structures



Ideas on data management



Different kinds of data involved

- Input data
- Initial structures (files from which simulations can be started)
- Raw simulation data
- Processed simulation results
- (Simulation codes)

Input data

- Data type
 - Gibbs energy, diffusion coefficient/mobility, interface energy grain boundary energy
- Specifications
 - Temperature, composition, orientation, ...
- Value
- Source
 - Reference, method
- Accuracy

Initial microstructure

- Description
 - Number of components, grains, phases, system size, ...
- Source
 - Method / how is the structure obtained, eg experimental micrograph, voronoi tessellation,
- Data
 - Files with the values of the phase-field variables as a function of position
 - Important to specify the format well

Raw simulation data

- File with (model) input data
 - Values + references
 - Or reference to 'input data' files
- Reference to initial structure files
- Numerical details
 - Discretization technique, name+version simulation program, time stepping, grid spacing
- Data
 - Files with the values of the phase-field variables as a function of position and time
 - Important to specify the format well

Processed simulation results

- Type
 - Images, movies, graphs, ...
- Content
 - Eg mol fraction B as a function of time, grain size distribution, ...
- Reference to the raw data files, input files, initial data files

